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# Initial Characterization of the Prepared Au-Decorated TiO<sub>2</sub>:Fullerene Films Using Electro spray Method

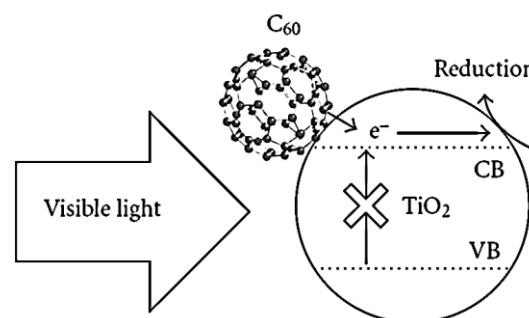
*This study aims to enhance the properties of fullerene composite films and investigate the impact of incorporating TiO<sub>2</sub> on their structural, topographic, and optical characteristics. A hybrid method was employed to prepare these composite films by combining laser ablation to produce the colloidal of carbon, titanium dioxide and gold, then using the electro-spraying technique to produce TiO<sub>2</sub>:fullerene films and then decorate them with gold nanoparticles. Addition of TiO<sub>2</sub> and decoration by gold have led to variations in the structural phases and a decrease in both crystallite and particle sizes. Similar effects were observed for band gap energy and Urbach energy.*

**Keyword:** Titanium dioxide; Fullerene; Electro-spray; Wasted batteries; Laser ablation  
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## 1. Introduction

Carbon-based nanomaterials, such as graphene, fullerenes, carbon nanotubes, and carbon dots, have been extensively studied [1]. One of fullerenes is C<sub>60</sub> also known as buckminsterfullerene or simply fullerene, is a carbon-based nanomaterial with consisting of 60 carbon atoms arranged in a hollow sphere [2]. Due to its unique a unique structural, electronic, and optical properties [3], involves many applications including photo-optical, supercapacitors, biomedical [4], drug delivery [1] applications. Addition to transistors and solar cells [5]. Titanium oxide (TiO<sub>2</sub>) in one-dimensional (1D) nanostructure has a large surface/volume, a volume of the pore, and a mean-free-path length of electrons comparable with the two directions perpendicular to the long axis of the particles [5-9]. Therefore, TiO<sub>2</sub> with these properties is attractive and promising to the high-activity TiO<sub>2</sub>-based photocatalyst [10-13]. Despite the relatively large gap in TiO<sub>2</sub> (more than a volt), the presence of C<sub>60</sub> provides a smooth transfer of electrons in the presence of ultraviolet radiation, as it is in the presence of the non-local P structure that acts as a good acceptor for photocatalyzed electrons. In the C<sub>60</sub>/TiO<sub>2</sub> system, the conduction band of TiO<sub>2</sub> is higher compared to that of C<sub>60</sub>/C<sub>60</sub><sup>-</sup> (vs. NHE), so photoelectrons can transfer from TiO<sub>2</sub> nanoparticles to C<sub>60</sub> in the presence of UV light [14]. Moreover, the amalgamation of TiO<sub>2</sub> and C<sub>60</sub> results in heightened absorption attributed to an increase in surface electric charge. This, in combination with the heightened Schottky barrier between the deposited noble metal and the electron catalyst, serves to facilitate the capture of electrons by the noble metal. Consequently, it impedes the recombination of electron-hole pairs and amplifies the interaction for interfacial charge transfer. In this perspective, TiO<sub>2</sub> and gold decoration are expected to improve the

optical properties of fullerenes towards photocatalysis [15]. Figure (1) explains the electronic reaction between C<sub>60</sub> and TiO<sub>2</sub>.



**Fig. (1) A diagram of the photoelectron transfer model of TiO<sub>2</sub>-C<sub>60</sub> composite by visible light effect [18]**

The photocatalytic activity enhancement of C<sub>60</sub> and some compounds by C<sub>60</sub> were discussed by many studies, such as Yu et al. [14], who have synthesized fullerene-modified TiO<sub>2</sub> nanocomposites and studied the enhancement of photocatalytic activity of the C<sub>60</sub>/TiO<sub>2</sub> nanocomposites. Dai et al. [16] have improved the photocatalytic of the modifying PbMoO<sub>4</sub> by C<sub>60</sub>. Tahir et al. [17] have fabricated fullerene-WO<sub>3</sub> composite systems for the photocatalytic. Also, Shahzad et al. [18] have fabricated heterojunction composites from WO<sub>3</sub>/fullerene/Ni<sub>3</sub>B/Ni(OH)<sub>2</sub> structures. Peng [19] has reported a theoretical study on the ability of using a Monolayer Fullerene Networks (MLFN) as photocatalyst for water treatment. In 2022, Fadhil et al. have studied the photocatalytic behaviour of C<sub>60</sub>-TiO<sub>2</sub>-ZnO, C<sub>60</sub>-ZnO, C<sub>60</sub>-TiO<sub>2</sub> nanocomposites. Liping et al. [20] have enhanced it by creating C<sub>60</sub>@NU-901. Likewise, Munawar et al. [21] have fabricated a dual-functional La<sub>2</sub>O<sub>3</sub>-C<sub>60</sub>

nanocomposite as an effective photocatalyst and supercapacitors electrode.

In this study, the structural, morphological, and optical properties, of Au-decorated TiO<sub>2</sub>:fullerene films were investigated as a proposed photocatalyst.

## 2. Materials and Method

Carbon pellet extracted from battery electrodes were prepared by pressing carbon powders by means of a hydraulic press [22]. The same applies to TiO<sub>2</sub> powder to fabricate a TiO<sub>2</sub> pellet, which was placed in 50 ml of deionized water (DW), then bombarded with 3000 shots from pulsed Nd:YAG laser of 1064 nm wavelength and 100 mJ energy. The duration, width, and intensity of laser at the spot of shot were 6 Hz, 10 nm, and 4 J.cm<sup>-2</sup>, respectively, and the distance between the surface target and the laser source was 12 cm. Then, carbon/TiO<sub>2</sub> colloidal was prepared by irradiating the carbon pellet in colloidal TiO<sub>2</sub> in 10, 30, and 50 vol.% of 50 ml of DW using the same laser parameters as well as the same number of laser shots. Finally, for decoration, the prepared films were sprayed by gold nanoparticles (AuNPs) colloidal, which was prepared with the same previous condition, but 1000 laser shots.

Carbon colloidal of 50 ml was sprayed on quartz and Au-coated glass substrates at 300±25 °C by electro spray technique with dc electrical bias potential of 5 kV, flow rate about 1 ml/min, ON/OFF spraying time of 10 and 30s, respectively. Figure (2) shows the schematic of the electro spray device used in this work.

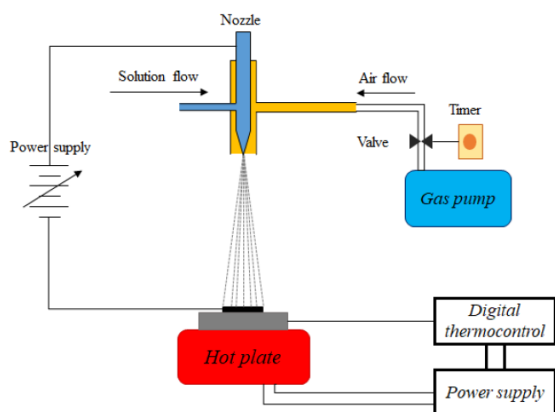


Fig. (2) Schematic of the electro spray device used in this work

The structural properties of the prepared films were performed on x-ray diffractometer (XRD) using CuK $\alpha$  radiation and accelerated voltage of 40 kV. The crystallite size had been estimated based on the Williamson-Hall relation [23]

$$D_{hkl} = \left( \frac{A\lambda}{\cos\theta \cdot \beta_{hkl}} \right) + (4\epsilon \sin\theta) \quad (1)$$

where  $D$  (nm) is the crystallite size,  $A$  is a dimensionless shape factor (0.89),  $\lambda$  is the wavelength of the x-ray beam (nm),  $\beta_{hkl}$  is the full-width at half maximum (FWHM) of the diffraction peaks,  $\epsilon$  is the strain, and  $\theta$  (rad) is the diffraction angle

The quantitate phase measurements were achieved using the reference intensity ratio (RIR) method [22]. A field-emission scanning electron microscope (FE-SEM, INSPECT-550) was used to introduce the surface morphology of the prepared films. The optical properties of the prepared films were investigated using a PG-T-80 UV-Visible spectrophotometer in the spectral range 200-900 nm.

## 3. Result and Discussion

The XRD patterns of the prepared films have been analyzed based on ICSD 98-060-2518, ICSD 98-005-6668, ICSD 98-009-5370, ICSD 98-007-5506, ICSD 98-000-9161, ICDD 98-005-6902, and ICSD 98-005-6909 cards for C<sub>60</sub>, C<sub>60</sub>-Polymeric, C<sub>70</sub>, TiO<sub>2</sub> (Rutile), C<sub>14</sub>H<sub>10</sub> (Ravatite), and C<sub>20</sub>H<sub>36</sub> (Dinitite). The thumbnail at the top of Fig. (3) represents the real XRD pattern of pure fullerene, in which the amorphous phase is hidden. Figure (3) shows XRD patterns of the prepared fullerenes films. XRD pattern of the prepared carbon films appears to be polycrystalline of C<sub>60</sub> in (211), (311), (146), (352), and (651), where the dominant phase was (311) at 20.77° with crystallite size of about 80.76 nm, in conjunction with two structures: C<sub>60</sub>-polymer (333), (170), and (600), and both (420) and (307) for C<sub>70</sub>. It is clear that the addition of TiO<sub>2</sub> has led to significant changes in the structures, as shown in diffraction patterns. There is instability in the dominant phase site with variation of TiO<sub>2</sub> ratio and structures with the adding of TiO<sub>2</sub> [24]. At TiO<sub>2</sub> of 10 vol.%, the C<sub>60</sub>-polymer ratio increased significantly with emergence of C<sub>14</sub>H<sub>10</sub> phase (102). This structure is possibly a result of the dissociation of one of the fullerene molecules because of the high temperature and pressure in the plasma bubble generated by the laser in the presence of hydrogen resulting from the instantaneous dissociation of water with the same reason. The dominant phase is (600) which related to C<sub>70</sub> with crystallite size of about 69 nm with a decrease in the quantity ratio to C<sub>60</sub> as shown in table (1). With TiO<sub>2</sub> of 30 vol.%, it was observed that the polymerization continued to dominate the structure and the dominating phase returned to C<sub>60</sub> (221) at 20.81° with crystallite size of about 31.68 nm and an increase of both RIR of C<sub>60</sub>-Polymeric and C<sub>70</sub> and vanishing of C<sub>14</sub>H<sub>10</sub> phase. It is interesting that when the percentage of TiO<sub>2</sub> was increased to 50 vol.%, a phase emersion of Bis(cyclopentadienyl)bis(m-methoxybenzoato) titanium(IV) Ti<sub>4</sub>C<sub>104</sub>O<sub>24</sub> was observed. This result confirms the reaction of the Ti<sup>+</sup> ion with carbon atoms and O<sup>-</sup> ion. The dominate phase was related C<sub>70</sub> (141) at 21.60° in conjunction a drastic drop in RIR and an increase for C<sub>60</sub> and C<sub>70</sub>, respectively. Moreover, no peaks were seen for AuNPs due to their small amount.

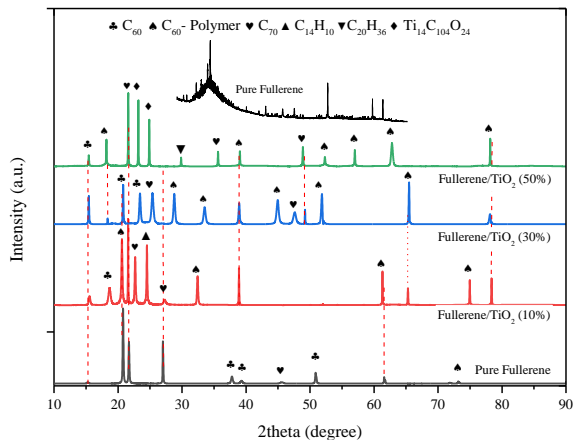


Fig. (3) XRD patterns of the prepared fullerenes and Au-decorated TiO<sub>2</sub>:fullerene films

Table (1) Influence of TiO<sub>2</sub> addition on the RIR of the prepared Au-decorated TiO<sub>2</sub>: fullerene films

Ratio of TiO <sub>2</sub> (vol.%)	C <sub>60</sub> (%)	C <sub>60</sub> -Polymeric (%)	C <sub>70</sub> (%)
0	48.4700	5.51	46.02
10	6.7400	22.42	35.54
30	13.8800	70.64	15.48
50	2.9315	31.68	30.32

Figure (4) represents images of the prepared fullerenes and Au-decorated TiO<sub>2</sub>:fullerene films. Fullerene films showed a variety of different shapes of nanoparticles, it is remarkable that there is a type of separation between the different particles, especially the rods. These nanorods can be C<sub>60</sub>-polymeric with an average diameter of 39 nm, as shown in Fig. (4a). The separation between the particles and the inhomogeneity of the distribution can be attributed to the segregation of impurity phenomena as a result of the difference in the surface energy of the different particles [25]. Figure (4b) shows a better distinction for polymeric C<sub>60</sub> with a change in the form of the formed particles that appear like disks longitudinally stacked (about 80 nm of thickness) tablets interspersed with spherical particles (about 34 nm of size) as they approach the assembly line. Increasing of TiO<sub>2</sub> ratio to 30 and 50 vol.% has led to appearance of material as heterogeneous aggregates with undifferentiated components as shown in figures (4c) and (4d). For clarity, the thumbnail images have been processed at a lower magnification to show the general distribution pattern of the deposited films. It appears that there is an evolution of the texture, where becomes more regular (as a fiber structure) with increasing TiO<sub>2</sub> ratio and begins to lose its agglomerated state in the form of fibrils when increasing to 50 vol.%. The cross section images of films showed a change in the deposited thickness of films to be 1172±664, 1373±217, 579±238.8, and 540±199.8 nm for 0, 10, 30, and 50 vol.% of TiO<sub>2</sub> ratio, respectively, as shown in Fig. (4e, f, g, and h).

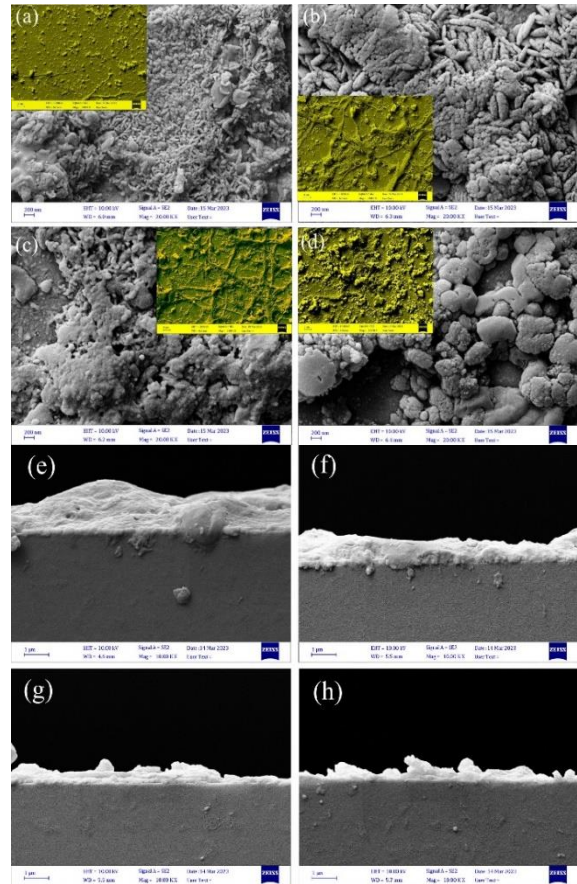


Fig. (4) FE-SEM images of the prepared Au-decorated TiO<sub>2</sub>:fullerene films (a, b, c, and d) for the surface, (e, f, g, and h) for thickness

Figure (5) illustrates the behavior of the optical absorption of the prepared films. Fullerene displays a very interesting, complex and characteristic absorption spectrum with UV light because it has 30 weakly conjugated double bonds. Fullerene shows two intense bands at about 210 and 275 nm. The other absorption band of fullerene appearing in the UV region is located at about 330 nm. In the visible region, the spectrum of fullerene is dominated by a band at about 404 nm and by a broad and relatively weak groups of band between 430 and 670 nm with subfeatures at 500, 530, 570, 600, and 628 nm [26-29]. Clearly, at 210 nm, there is relatively high absorption with decreasing and shift by adding of TiO<sub>2</sub>, blue shift for 30 and 50 vol.% and red for 10 vol.%, while at 385 nm for 0 vol.% TiO<sub>2</sub>, which is related to a peak of 410 nm (as previous studies mentioned [9]), the spectrum suffered from a blue shift due to containing other structural phases as shown in the XRD pattern in Fig. (3). Concurrently, adding TiO<sub>2</sub> has led this peak to expand and merge with the previous one for both 10 and 50 vol.%, while its prominence is clear at 30 vol.%, which may be related to the increase of C<sub>60</sub>-polymeric. There was also a clear decrease in the absorbance of films with a 30 vol.% TiO<sub>2</sub> ratio. The XRD pattern of fullerene:TiO<sub>2</sub> (30 vol.%) showed the absence of any

structures outside the fullerene structures, which led to the abundance of TiO<sub>2</sub> without linkage that caused dominance of the optical TiO<sub>2</sub> behavior in addition to heavy attendance of C<sub>60</sub>-polymeric [18]. Although the peaks of 500 and 530 nm (which is believed to be a contribution from the localized surface plasmon resonance (LRSP) of AuNPs) did not appear in the reference films as they became transparent when adding TiO<sub>2</sub> at all ratios with a clear convexity in the spectrum within the 550 nm regions down. This appearance can be confirmed as it represents the plasmon peak of AuNPs added as a decoration.

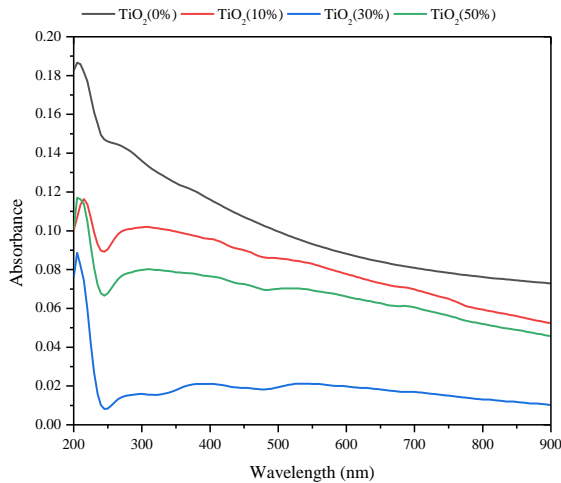


Fig. (5) Absorption spectra of the prepared Au-decorated TiO<sub>2</sub>:fullerene films with different ratios of TiO<sub>2</sub>

The bandgap energy ( $E_g$ ) can be estimated by the Tauc's equation as [28]

$$\alpha h\nu = A(h\nu - E_g)^n \quad (2)$$

where  $h\nu$  is the photon energy,  $\alpha$  is the absorption coefficient,  $A$  is a constant, and  $n$  has the values of 1/2 and 2/3 for direct and indirect transitions, respectively

The plots of  $(\alpha h\nu)^{1/2}$  versus  $h\nu$  are depicted in Fig. (6) for all samples. Increasing TiO<sub>2</sub> ratio has clearly influenced the values of bandgap of the prepared Au-decorated TiO<sub>2</sub>:fullerene films, as the bandgap was decreased with increasing TiO<sub>2</sub> ratio. Although, the energy gaps of C<sub>60</sub> and C<sub>70</sub> obtained from this study are large when compared to the typical values (1.86 and 1.57 eV). The lack of complete purity of the prepared films (containing amorphous phases) as proven by the analysis of the XRD patterns (Fig. 3), will lead to a large difference in the values of the energy gap and the mechanisms used by the material in the optoelectronic transitions, as exploded in table (2).

Table (2) Summary of influence of TiO<sub>2</sub> addition on the bandgap and Urbach energies of the prepared Au-decorated TiO<sub>2</sub>:fullerene films

Ratio of TiO <sub>2</sub> (vol. %)	E <sub>g</sub> (eV)	E <sub>u</sub> (eV)	Crystallite Size (nm)	Lattice Strain [%]
0	2.15	3.60	80.74	0.265
10	1.87	1.36	69.02	0.298
30	1.63	1.42	31.68	0.213
50	1.62	1.42	96.69	1.845

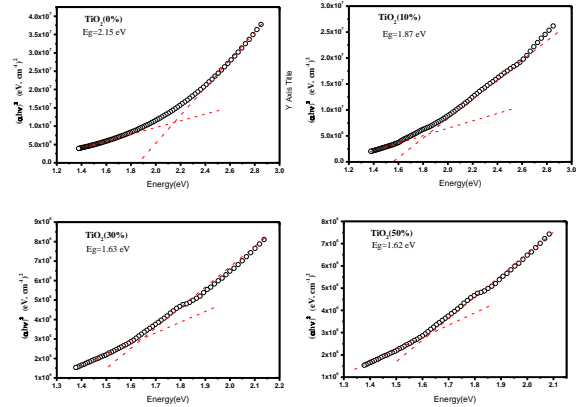


Fig. (6) The optical bandgap of the prepared Au-decorated TiO<sub>2</sub>:fullerene films with different TiO<sub>2</sub> ratios

Urbach energy ( $E_u$ ) related to both band tails of the localized states with the microstructural lattice disorders, and the crystal defects, can be estimated by the following [29]

$$\alpha = \alpha_0 \exp\left(\frac{h\nu}{E_u}\right) \quad (3)$$

Figure (7) is a graphical representation of Eq. (3) for the prepared films with varying TiO<sub>2</sub> ratio, where  $E_u$  or the disturbance can be estimated by calculating the reverse slope of the linearly fitted line. However,  $E_u$  decreases with increasing TiO<sub>2</sub> ratio but keeps constant with 30 and 50 vol.%. Smaller strain decreases the structural disorder since it will influence the built-in electric field at the grain boundary and hence will consequently inhibit a disturbance of the density of the states [29,30]. Referring to the calculations derived from the analysis of the XRD patterns, it is noted that there is a correlation between the values of the Urbach energy and the strain values for the ratios of 0, 10, and 30 vol.%.

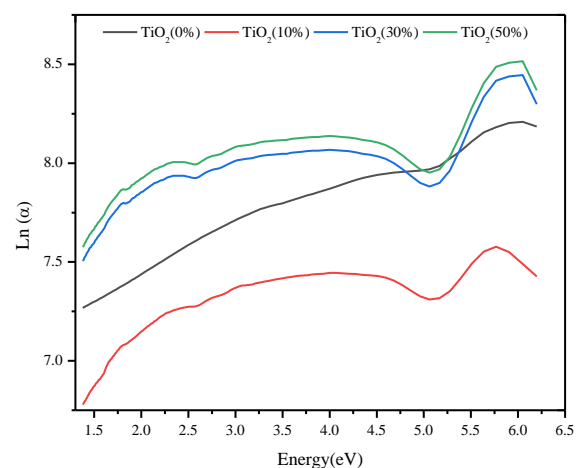


Fig. (7) Urbach energies of the prepared Au-decorated TiO<sub>2</sub>:fullerene films as functions of energy

#### 4. Conclusion

Au-decorated TiO<sub>2</sub>:fullerene films were successfully prepared to produce fullerene and TiO<sub>2</sub>:fullerene composites. The wasted batteries

were used as a source for the pure carbon. In general, structural, morphological, and optical properties were affected by the variations in TiO<sub>2</sub> ratio in the prepared composites mainly on the crystalline phases, surface formation, and optical properties. These results encourage the selection of dry battery electrodes as a raw source for fullerenes and the Au-decorated TiO<sub>2</sub>:fullerene films in several applications, such as photodetectors, gaseous and biosensors based on photocatalytic activity of such films.

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