The influence of partial substation of antimony & lanthanum oxides on electrical and structural properties for the superconductor compound Bi<sub>2-x</sub>Sb<sub>x</sub>Ba<sub>2</sub>Ca<sub>2-</sub> yLayCu<sub>3</sub>O<sub>10+δ</sub>

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Lock-in Amplifiers up to 600 MHz





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# The Influence of Partial Substation of Antimony & Lanthanum Oxides on Electrical and Structural Properties for the Superconductor Compound Bi<sub>2-</sub> <sub>x</sub>Sb<sub>x</sub>Ba<sub>2</sub>Ca<sub>2-y</sub>La<sub>y</sub>Cu<sub>3</sub>O<sub>10+δ</sub>

Laheeb A. Mohammed<sup>1,a)</sup>, Haider S. Hussein<sup>2</sup>, Haider M.J.Haider<sup>3</sup>, Kareem A. Jasim<sup>1</sup>, Auday H. Shaban<sup>1</sup>, Samir G. M. Askar<sup>1</sup>, Fouad W. Ali<sup>4</sup>

<sup>1</sup> Department of Physics, College of Education for Pure Science Ibn-Al-Haitham, University of Baghdad, Baghdad, Iraq <sup>2</sup>General Directorate of Karbala Education, Iraq <sup>3</sup>University of kufa\_Faculty of Education for Girls \_Physics Department, Iraq <sup>4</sup> Ministry of Education, Baghdad, Iraq. <sup>a</sup>Laheebfrahmed@yahoo.com

**Abstract.** In this research the  $Bi_{2-x}Sb_xBa_2Ca_{2-y}La_yCu_3O_{10+\delta}$  compound was prepared and studied the influence of partial substation of  $Sb_2O_3$  and  $La_2O_3$  on the electrical and structural properties Samples synthesis by solid state reaction method. The examination of XRD diffraction shown that all samples have bulk polycrystalline with orthorhombic structure. The results shows that the increasing of the c-axis lattice constant for the samples substation with  $Sb_2O_3$  and  $La_2O_3$  as compared with those have no content, It was found that changing in lattice parameters (a,b), ratio c/a ,mass density and volume fraction Vphases with increases of  $(Sb_2O_3,La_2O_3)$  concentration. All samples have highest phase 2223 major relative to the other phases. The electrical properties tested by using four probes technique to calculation critical temperatures, it was found the sample  $Bi_{1.6}Sb_{0.4}Ba_2Ca_{1.8}La_{0.2}Cu_3O_{10+\delta}$  has a highest critical temperatures Tc=122.5K.

Keywords: critical temperatures, superconductor, X-ray diffraction, structural properties, electrical properties

## **INTRODUCTION**

Superconductivity, is described as a phase of transition of electrons, in which the conductor loss all electric current resistance under particular circumstances and shows diamagnetism of complete.

Superconductivity is the phenomenon of complete lack of electrical resistance in specific materials when they cooled below appointed temperature called transition temperature or critical (Tc) which differs from material to material another [1-3]. Superconductors of copper oxide are the extreme important high critical temperature superconductors. The finding of a room temperature (300 K) superconductor must activate a rampage of large technological. There are claims of a superconductor making at (300 K) (for example, see, www.superconductors.org, 2011). But these claims are unaccepted by the community of scientific. Generally, it is accepted in the literature of scientific that the highest critical temperature is nearly up to 135 K at (760 mmHg) in the Hg-Ba-Ca-Cu-O system (Cantoni & Schilling, 1993). Though, critical temperature in this system can be increased to 180 K by external pressures of high.

We trust that the room temperature superconductor (300 K) discovery would be likely only when the microscopic mechanisms of oxide superconductors clarified [4].

Yet, up to date, the responsible of microscopic mechanisms for high critical temperature superconductivity are vague. In a modern article (Luiz, 2010), we have explain a simple manner to microscopic mechanisms of study in high critical temperature superconductors. It is well recognized that

Technologies and Materials for Renewable Energy, Environment and Sustainability AIP Conf. Proc. 2190, 020018-1–020018-6; https://doi.org/10.1063/1.5138504 Published by AIP Publishing. 978-0-7354-1937-7/\$30.00 the superconducting state is described by a quantum macroscopic state that get up from a condensation of Bose-Einstein of paired electrons (pairs of Cooper)[4].

The discovery of high-TC superconductors of oxide like Bi (Pb)-Sr-Ca- Cu-O (BSCCO) and Y-Ba-Cu-O (YBCO) systems of ceramic was the starting point of a new rampage in chemistry and physics and in technology activities of high. Generally most ceramic high critical temperature superconductors are formed by sintering given mixes of raw constituents. Yet, it is tricky to get ceramics complex forms with durability of mechanically high by the sintering. A poorly sintered materials with low density was produced by the traditional solid state reaction (SSR) method of preparing BSCCO system.

Bi-based is family of cuprate superconductor symbolize by the generic formula  $Bi_2Sr_2Ca_{n-1}Cu_nO_{2n+4}$  (n equal 1, 2, 3) (called Bi-2201, Bi-2212, and Bi-2223, correspondingly). The last two members are well-known to have remarkable properties for the essential research and uses in technology and industries [10-15].

## **EXPERIMENTAL WORK**

The Bi2-xSbx[Ba]\_2 [Ca]\_(2-y) [[[La]\_y Cu]\_3 O\_(10+ $\delta$ ) with (x=0.1,0.2,0.3,0.4,0.5 and y=0.1,0.15,0.2,0.25) compounds were these samples synthesized by execution of (SSR)method, the amounts powders required of pure oxides .the utilized oxides high purity is nearly pure (99.99%) of Bi2O<sub>3</sub>,Sb<sub>2</sub>O<sub>3</sub>,La<sub>2</sub>O<sub>3</sub> BaO,CaO and CuO.

The reactants were estimated by using a balance of sensitive, with sensitivity order (10-4) g. The reactants were mixed together by a gate mortar with isopropanol ( $C_2H_3O_5$ ) addition to be homogenize. To get rid of the water vapor the powders dried for 1.5h under 150 ° C, and Then grind and mixed by a 6h electric mixer of spiral for optimal homogenization and for precise powders. The resulting powder then compressed utilizing a piston of hydraulic for 2 minutes under pressure of 0.7G Pascal, in the shape of discs with a 0.75 cm radius and a 0.3-0.35 cm thickness. the samples sintering at 760°C for (160h) with average of (5°C per minute) to get a material of bonding and to guarantee optimum gradual diffusion between the atoms . Thereafter the samples were cooled at 5°C per minute (same heating rate) to room temperature. To obtain the structural properties of the samples. The samples were examined within the range of the diffraction angle (10-80). The constants (a, b, c) were calculated mathematically utilizing d-values and (Miller indices) reflections of the observed XRD pattern during the program of software In addition, the cell unit density was measured, and the phases formed percentages in the samples [16-17]. The fraction of volume of the phase calculation based on the formula of following [17].

$$\boldsymbol{Vph} = \frac{\Sigma I^{\circ}}{\Sigma I^{\circ} + \Sigma I 1 + \Sigma I 2} *100\% \dots (1)$$

P is the hole-carrier concentrations per Cu ion, It is calculated by means of the following relation [18]:

## **RESULT AND DISCUSSION:**

### **Structural properties**

The XRD diffraction of pure and other samples replacement of Sb<sub>2</sub>O<sub>3</sub>, La<sub>2</sub>O<sub>3</sub> were investigated at 2 $\Theta$  =( 10°- 80°), the results that shown in figure (1) were all polycrystalline with orthorhombic structure system and corresponded to the Bi-2223 phase. The x-ray diffraction patterns prove some impurities presence in very small concentration. fig (1) shows that there are two main phases in every specimens including the high critical temperature phase (2223) which is the dominant, the low critical temperature phase (2212) and impurities in little amount. The substitution of Bi by Sb<sub>2</sub>O<sub>3</sub> and Ca by La<sub>2</sub>O<sub>3</sub> causes shifting in diffraction angle 2 $\Theta$  very small to ward increment angle and change in the magnitude of intensities of peaks. The lattice parameters calculated show in table (1) noticed that the increase of substitution concentration causes increasing in value of c-axis due to variation in ionic radius. The parameters of lattice based on method of Cohen's least square, calculation by software program [17].

X	Y	a (A°)	b (A°)	c(A°)	c/a ratio
0	0	5.434	5.433	36.789	6.770151
0.1	0	5.4221	5.4227	36.899	6.805297
0.2	0.1	5.4201	5.4216	37.121	6.848767
0.3	0.15	5.4189	5.4198	37.266	6.877041
0.4	0.2	5.4178	5.4165	37.544	6.92975
0.5	0.25	5.411	5.4034	37.622	6.952874
v(A°)³	dm(g/cm <sup>3</sup> )	V ph(1223)%	V p <i>h</i> (1212)%	V ph(1201)%	Vp impurities%
<b>v(A°)³</b> 1086.119	dm(g/cm <sup>3</sup> ) 442.7493	<b>V ph(1223)%</b> 74.102	V ph(1212)% 8.826	<b>V ph(1201)%</b> 7.953	Vp impurities% 9.117
v(A°) <sup>3</sup> 1086.119 1084.92	dm(g/cm <sup>3</sup> ) 442.7493 436.2161	V ph(1223)% 74.102 71.2753	V ph(1212)% 8.826 14.3902	<b>V ph(1201)%</b> 7.953 5.7435	<b>Vp impurities%</b> 9.117 5.0503
v(A°) <sup>3</sup> 1086.119 1084.92 1090.823	dm(g/cm <sup>3</sup> ) 442.7493 436.2161 751.1744	V ph(1223)% 74.102 71.2753 74.2033	V ph(1212)% 8.826 14.3902 10.0957	V ph(1201)% 7.953 5.7435 9.6064	<b>Vp impurities%</b> 9.117 5.0503 6.0944
v(A°) <sup>3</sup> 1086.119 1084.92 1090.823 1094.478	dm(g/cm <sup>3</sup> ) 442.7493 436.2161 751.1744 904.1952	V ph(1223)% 74.102 71.2753 74.2033 76.395	V ph(1212)% 8.826 14.3902 10.0957 13.003	V ph(1201)% 7.953 5.7435 9.6064 5.865	<b>Vp impurities%</b> 9.117 5.0503 6.0944 4.734
v(A°) <sup>3</sup> 1086.119 1084.92 1090.823 1094.478 1101.748	dm(g/cm <sup>3</sup> ) 442.7493 436.2161 751.1744 904.1952 1052.24	V ph(1223)% 74.102 71.2753 74.2033 76.395 81.737	V ph(1212)% 8.826 14.3902 10.0957 13.003 12.7922	V ph(1201)% 7.953 5.7435 9.6064 5.865 8.3812	<b>Vp impurities%</b> 9.117 5.0503 6.0944 4.734 5.1461

 $\begin{array}{l} \textbf{Table .1. Values of lattice parameter a, b, c, c/a, $\rho_M,V,V_{phase}$ of $Bi_{2-x}Sb_xBa_2Ca_{2-y}La_yCu_3O_{10+\delta}$ compound $(x=0.1,0.2,0.3,0.4,0.5 y=0.1,0.15,0.2,0.25)$} \end{array}$ 



Figure.1. X-ray Diffraction pattern of  $Bi_{2-x}Sb_xBa_2Ca_{2-y}La_yCu_3O_{10+\delta}$  compound (x=0.1, 0.2, 0.3, 0.4, 0.5 y=0.1, 0.15, 0.2, 0.25)

In the proportional intensities comparison of samples X-ray Diffraction patterns utilizing  $Sb_2O_3 = 0.0, 0.1, 0.2, 0.3, 0.4, 0.5$  and y=00, 0.05, 0.1, 0.15, 0.2, 0.25.

Figure. 1 shows that every have reflective intensity of high phase (H-peaks) reflections and low phase (L-peaks) reflections and lower critical temperature reduced by Sb. The high-critical temperature phase reflections of the free specimen Sb, La = 0 have a decrease intensity than those containing Sb, La. The lattice parameters (a, b, c), density of mass ( $\rho$ M) and fraction of volume (Vphase) revealed in Table (1), It was clear that the increasing of the Sb,La concentrations of every our specimens yield variations on the lattice parameter values, c/a,  $\rho$ M and volume of unit cell.

#### **Electrical properties**

We examined the electrical properties for the samples by the four probe technique. Difference the temperature with the resistance as synthesized  $Bi_{2-x}Sb_x Ba_2Ca_{2-y}La_yCu_3O_{10+\delta}$  (with y=0.0,0.05,0.1,0.15,0.2,0.25 and x=0.0,0.1,0.2,0.3,0.4,0.5). Resistance of natural to every mineral specimens like metal behavior with regard to temperature. Fig. (2) Demonstrate a measured pattern of temperature versus resistivity (T- $\rho$ ) with altered concentrations.



Figure.2. The resistivity as function of temperature of Bi<sub>2-x</sub>Sb<sub>x</sub>Ba<sub>2</sub>Ca<sub>2-y</sub>La<sub>y</sub>Cu<sub>3</sub>O<sub>10+δ</sub> compound (x=0.1, 0.2, 0.3, 0.4, 0.5 y=0.1, 0.15, 0.2, 0.25)

transition temperature values (T<sub>c</sub>) for  $Bi_2Ba_2Ca_2Cu_3O_{10+\delta}$ ,  $Bi_{1.9}Sb_{0.1}Ba_2Ca_2Cu_3O_{10+\delta}$ ,  $Bi_{1.8}Sb_{0.2}Ba_2Ca_{1.9}La_{0.1}Cu_3O_{10+\delta}$ ,  $Bi_{1.7}Sb_{0.3}Ba_2Ca_{1.85}La_{0.15}Cu_3O_{10+\delta}$ ,  $Bi_{1.6}Sb_{0.4}Ba_2Ca_{1.85}La_{0.2}Cu_3O_{10+\delta}$ phases are 101.6 K, 113 K, 118 K, 119.5 and 122.2 K respectively we can observed growth in degree of critical temperature values (T<sub>c</sub>), then be go down in concentration of x=0.5,y=0.25  $Bi_{1.5}Sb_{0.5}Ba_2Ca_{1.8}La_{0.2}SCu_3O_{10+\delta}$  equal to 117.5K .Can be listed of categorically that specimens under this labor are "a good quality", where the optimal extreme value of transition temperature is found to be concentration  $Bi_{1.6}Sb_{0.4}Ba_2Ca_{1.85}La_{0.2}Cu_3O_{10+\delta}$  resemble to the doped hole optimal level shows in table (2)

 $\label{eq:table2.Transition temperature $T_{c(onset)}$, $(T_{c(offset)}$, and Hole concentration of $B_{12-x}Sb_xBa_2Ca_{2-y}La_yCu_3O_{10+\delta}$ compound $(x=0.1,0.2,0.3,0.4,0.5 y=0.1,0.15,0.2,0.25)$}$ 

			,,	··-,··,··,	
X	Y	T <sub>c(ON)</sub> K	T <sub>c(OFF)</sub> K	T <sub>c</sub> K	P (Hole) concentration
0	0	105.2	98	101.6	0.114552
0.1	0	122	104	113	0.129359
0.2	0.1	126	110	118	0.138911
0.3	0.15	127	112	119.5	0.142781
0.4	0.2	127.5	117	122.5	0.154555
0.5	0.25	120	115	117.5	0.137771

The AFM images show a uniform granular surface morphology. The surface morphology of compound was illustrated in the figure (3). It can be seen that the morphology of surface results appeared that the surface roughness increased, the increases could be possibly due to the increasing of La<sub>2</sub>O<sub>3</sub>,Sb<sub>2</sub>O<sub>3</sub> concentration and time of sintering.



Figure 3. Reveals the (3-D) AFM images f surface morphology of Bi<sub>2-x</sub>Sb<sub>x</sub>Ba<sub>2</sub>Ca<sub>2-y</sub>La<sub>y</sub>Cu<sub>3</sub>O<sub>10+8</sub> compound (x=0.1, 0.2, 0.3, 0.4, 0.5 y=0.1, 0.15, 0.2, 0.25)

### CONCLUSION

In our work prepared compound superconductor  $Bi_{2-x}Sb_xBa_2Ca_{2-y}La_yCu_3O_{10+\delta}$  where x=0.1, 0.2, 0.3, 0.4, 0.5 and y=0.1, 0.15, 0.2, 0.25 using solid state reaction method and structural electrical properties examination. The results of XRD diffraction test appeared that the compound had polycrystalline with a orthorhombic structure. The transition temperature (T<sub>c</sub>) of the grown samples up to x=0.4,y=0.2 have been have observe that maximum T<sub>c</sub> 122.2K. In all samples The critical temperature were sensitive to the Sb, La concentrations. Transition temperature and volume fraction (V<sub>phase</sub>) increment and holes concentration with increase (Sb,La )concentration

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